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# The New Superconductors

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Table 8.1. Progress in Raising the Superconducting Transition Temperature  $T_c$ Since the Discovery of Cuprates in 1986

Material	$T_{\rm c}\left({\rm K}\right)$	Year
Ba <sub>x</sub> La <sub>5-x</sub> Cu <sub>5</sub> O <sub>9</sub>	30-35	1986
$(La_{0.9}Ba_{0.1})_2Cu_4O_{4-x}$ (at 1-GPa pressure) <sup>a</sup>	52	1986
YBa <sub>2</sub> Cu <sub>3</sub> O <sub>7-x</sub>	95	1987
Bi <sub>2</sub> Sr <sub>2</sub> Ca <sub>2</sub> Cu <sub>3</sub> O <sub>10</sub>	110	. 1988
Tl <sub>2</sub> Ba <sub>2</sub> Ca <sub>2</sub> Cu <sub>3</sub> O <sub>10</sub>	125	1988
Tl <sub>2</sub> Ba <sub>2</sub> Ca <sub>2</sub> Cu <sub>3</sub> O <sub>10</sub> (at 7-GPa pressure)	131	1993
HgBa <sub>2</sub> Ca <sub>2</sub> Cu <sub>3</sub> O <sub>8+x</sub>	133	1993
HgBa <sub>2</sub> Ca <sub>2</sub> Cu <sub>3</sub> O <sub>10</sub> (at 30-GPa pressure)	147	1994

<sup>&</sup>lt;sup>a</sup>A pressure of 1 GPa is about 10,000 atm.

While this increase in  $T_{\rm c}$  itself is an amazing result, a high-transition temperature is not the only property required to make new compounds useful for applications. For example if materials are to be used as wires in magnets, they must be malleable and ductile rather than brittle; in addition they must have high critical currents in large magnetic fields. Critical currents as high as those in niobium-tin have not yet been achieved in forms of the new materials that can easily be made into wires, although there are reports of comparable values in thin films on various substrates.

The Holy Grail that is being sought is a transition temperature much above room temperature. We say much above because devices must operate significantly below the transition  $T_c$  so that the critical current  $J_c$  and critical magnetic field  $B_c$  are sufficiently high. Very close to the transition temperature, the critical magnetic field is usually quite small, but we see from Figs. 3.4 and 3.5 that  $B_c$  and  $J_c$  continuously increase as the temperature is lowered below  $T_c$ . We need an operating temperature far below the critical surface in Fig. 3.15 so that both  $B_c$  and  $J_c$  are sufficiently large for the desired application.

#### 8.3. LAYERED STRUCTURE OF THE CUPRATES

All cuprate superconductors have the layered structure shown in Fig. 8.1: The flow of supercurrent takes place in conduction layers, and binding layers support and hold together the conduction layers. Conduction layers contain copper-oxide  $(CuO_2)$  planes of the type shown in Fig. 8.2; each copper ion  $(Cu^{2+})$  is surrounded by four oxygen ions  $(O^{2-})$ . These planes are held together in the structure by calcium  $(Ca^{2+})$  ions located between them, as indicated in Fig. 8.3. An exception to this is the yttrium compound in which the intervening ions are the element yttrium  $(Y^{3+})$  instead of calcium. These  $CuO_2$  planes are very close to being flat. In the normal state above  $T_{c+}$  conduction electrons released by copper atoms move about on these

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Figure 8.1. Layering schem layers for different sequence for several cuprates.

Figure 8.2. Arrangement of in a CuO<sub>2</sub> plane of the condu

Transition Temperature  $T_c$  n 1986

 Year	
1986	_
1986	
1987	
. 1988	
1988	
1993	
1993	
 1994	

t, a high-transition temperanpounds useful for applicas in magnets, they must be ney must have high critical igh as those in niobium-tin als that can easily be made ues in thin films on various

n temperature much above must operate significantly d critical magnetic field  $B_c$  ature, the critical magnetic .4 and 3.5 that  $B_c$  and  $J_c$  w  $T_c$ . We need an operating so that both  $B_c$  and  $J_c$  are

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ure shown in Fig. 8.1: The nd binding layers support yers contain copper-oxide r ion (Cu<sup>2+</sup>) is surrounded in the structure by calcium .3. An exception to this is the element yttrium (Y<sup>3+</sup>) being flat. In the normal toms move about on these

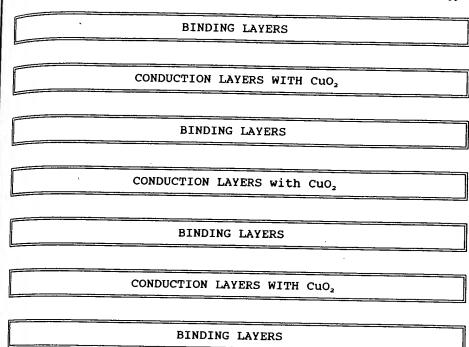


Figure 8.1. Layering scheme of the cuprate superconductors. Figure 8.3 shows details of the conduction layers for different sequences of copper oxide planes, and Fig. 8.4 presents details of the binding layers for several cuprates.

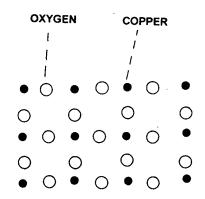


Figure 8.2. Arrangement of copper and oxygen atoms in a  $\text{CuO}_2$  plane of the conduction layer.

100		CHAPTER 8
	CuO <sub>2</sub>	
Conductio	n layer with one copper	oxide plane
	CuO <sub>2</sub>	
	Ca	Neody
	CuO <sub>2</sub>	
	n layer with two copper	
	CuO <sub>2</sub>	
	у	
	CuO <sub>2</sub>	
duction layer of	yttrium compound with t	wo copper oxide planes
	CuO <sub>2</sub>	
	Ca	Bism
	CuO <sub>2</sub>	
	Ca	
	CuO <sub>2</sub>	
0 - 1 - 1 - 1		
Figure 8.3. Conduction 1	layer with three copper ayers of the various cuprate superconductors induction layers of Fig. 8.1.	Thal
same electrons form	g electric current. In the supercondu the Cooper pairs that carry the super cuprate compound has its own specif	ercurrent in the planes.

mainly of sublayers of metal oxides MO, where M is a metal atom; Fig. 8.4 gives

the sequences of these sublayers for the principal cuprate compounds. These

binding layers are sometimes called charge reservoir layers because they contain

Figure 8.4. Sequences c metal ions. The parenthe:

Mercur

CHAPTER 8	LaO
	LaO
	Lanthanum Superconductor La₂CuO₄
r oxide plane	NAO
	NdO
	NdO
	Neodymium (electron) Superconductor Nd2CuO4
	BaO
r oxide planes	Cu0
1 Oxide planes	BaO
	Yttrium Superconductor YBa₂Cu₃O <sub>7</sub>
	sr0
	BiO
two copper oxide planes	BiO
	Sr0
	Bismuth Superconductor Bi <sub>2</sub> Sr <sub>2</sub> Ca <sub>n-1</sub> Cu <sub>n</sub> O <sub>2n+4</sub>
	BaO
	T10
	TlO
au auide planes	BaO
per oxide planes  ctors showing sequences of CuO <sub>2</sub> and	Thallium Superconductor Tl <sub>2</sub> Ba <sub>2</sub> Ca <sub>n-1</sub> Cu <sub>n</sub> O <sub>2n+4</sub>
	BaO
T share	нд(О)
inducting state below $T_c$ , these	

supercurrent in the planes. ecific binding layer consisting

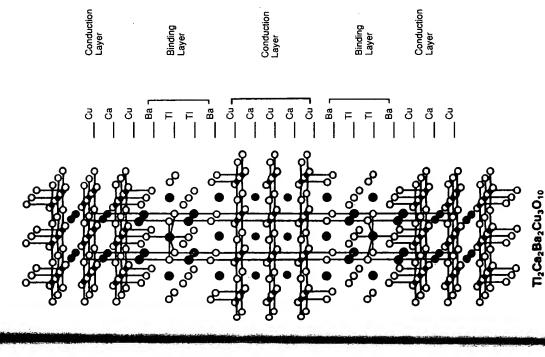
is a metal atom; Fig. 8.4 gives 1 cuprate compounds. These

ir layers because they contain

Figure 8.4. Sequences of MO sublayers in the binding layers of Fig. 8.1, where M stands for various metal ions. The parentheses around the oxygen atom O in the lowest panel indicates partial occupancy.

Mercury Superconductor  $HgBa_2Ca_{n-1}Cu_nO_{2n+2}$ 

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-0°

Figure 8.5. Layering schemes of three thallium compound superconductors  $\Pi_2 Ba_2 Ca_{n-1} Cu_n O_{2n+6}$  where there are  $n=1, 2, 3 CuO_2$  planes in the conduction layers, from left to right. [Adapted from Torardi et al., Science 240, 631 (1988).]

T2CaBa2Cu2O

Calcium Barium Copper Onygen

Thelium

T12Ba2CuO<sub>6</sub>

of randomly oriented grains. In he current flow capability of

La<sub>1-x</sub>,Sr<sub>x</sub>)<sub>2</sub>CuO<sub>4</sub> are hole-type erium-copper oxide, (Nd<sub>1-x</sub> trons rather than holes. The have trivalent positive ions:

(8.6)

(8.7)

itium (Sr2+) and cerium (Ce4+),

$$,CuO_4)$$
 (8.8)

$$l_2CuO_4$$
 (8.9)

one extra electron to form an contium subtracts one electron, iperconductor is hole-like. Any int both of these examples of lar, but not identical structures; cause most experiments are not

#### **!UCTURES**

ferred to as ceramics, they are erovskite refers to the particular eral perovskite, calcium titanate:) parts of the lanthanum comperovskite, with Cu present in 10t shown in Fig. 8.9) positions, Similarities between these two call La<sub>2</sub>CuO<sub>4</sub> a perovskite-type

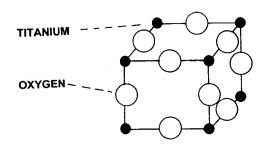


Figure 8.9. Sketch of the cubic unit cell of the mineral Perovskite, CaTiO<sub>3</sub>, showing titanium at the vertices and oxygen in the middle of the edges. Calcium, not shown, is in the center of the cube.

In contrast the ceramic designation is not based on structural grounds but on the similarity of the cuprate-superconducting compound and ceramic manufacturing process. For example La-Sr-Cu-O is made by heating mixtures of lanthanum oxide, strontium carbonate, and copper oxide in air at 900–1000 °C for 20 hours. Proportions of atoms in the initial mixture should be the same as in the end product, and for the compound  $(La_{0.9}Sr_{0.1})_2CuO_4$  the ratio La:Sr:Cu is 1.8:0.2:1. Materials are usually ground to a fine mixture before heating; after heating in air, they are cooled, pressed into pellets, and reheated from 900–1000 °C for several more hours.

We see in Fig. 8.10 that the superconductor  $(La_{1-x}Sr_x)_2CuO_4$  has only one copper oxide plane in its conduction layer and each copper ion is surrounded by

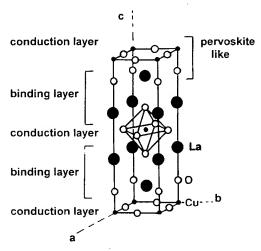


Figure 8.10. Atom positions in the tetragonal unit cell of the  $La_2CuO_4$  compound. When strontium is substituted for lanthanum in the superconducting compound  $(La_{1-x}Sr_x)_2CuO_4$  it replaces lanthanum in some of the La sites.

six neighboring oxygen ions; these form an 8-sided figure called an octahedron, as shown. The CuO<sub>6</sub> complex of one copper and six oxygens is present in all cuprate superconductors that have a single CuO<sub>2</sub> plane in their conduction layer. Figure 8.11 shows atom arrangements in the mercury compound HgBa<sub>2</sub>Ca<sub>2</sub>Cu<sub>3</sub>O<sub>10</sub>, which has three such planes in its conduction layer. In the upper and lower planes, copper ions have five neighboring oxygens forming a CuO<sub>5</sub> group with the shape of a pyramid, as shown. The middle copper ions have only four nearby oxygens, forming what is called a *square planar group* CuO<sub>4</sub>. If we consider removing the central copper oxide plane and one calcium layer from Fig. 8.11, we generate the two-plane structure in which all copper ions form CuO<sub>5</sub> pyramids. These structural details may somehow constitute important factors in determining why cuprates are such good superconductors.

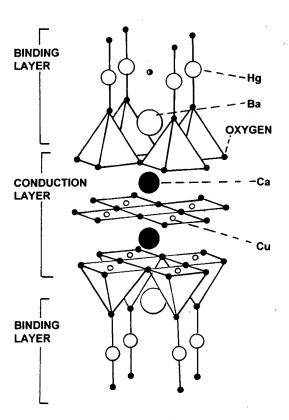


Figure 8.11. Atom positions in four unit cells of the superconducting compound  $HgBa_2Ca_2Cu_3O_{8tA}$  which has  $T_c = 133$  K. The copper ions of the upper  $CuO_2$  plane are hidden by the pyramids, and some partially occupied oxygen sites in the mercury Hg plane are not shown.

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#### 8.8. YTTRIU.

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Figure 8.12. First 1 Bednorz and K. A. l

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